

Imaging in Cryo-Electron Microscopy

Jaap Brink
JEOL USA, Inc.

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Outline

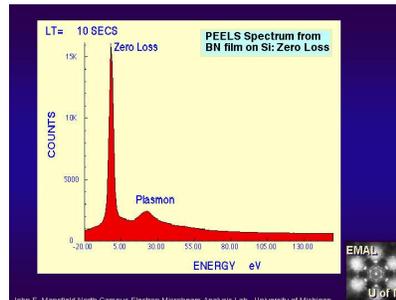
- Electron interactions
- Imaging theory
- Why are images not perfect?
- Analysis of cryo-EM data
- Future directions

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Electron Interactions with specimen

- Elastic
- Inelastic
- None



J. Mansfield, U Michigan

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Elastic interaction

- Mean free path – larger at higher kV
 - Single scattering event (kinematic)
- Angular dependency
 - Most of the electrons are scattered over large angles
 - Go either through the OL aperture (phase contrast) or are blocked (aperture contrast)
 - Without phase contrast, cryo-EM is not possible

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Inelastic scattering

- All electrons that lose energy while interacting with the specimen:
 - Absorbance (EELS)
 - X-rays (EDS)
 - Other electrons (Auger)
 - Other processes (phonons, plasmons)
 - Radiation damage

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Inelastic scattering I

- Beam damage:
 - Radiolysis
 - chemical bond breakage, free radical formation
 - loss of resolution
 - beam-induced motion
 - Knock-on damage
 - direct displacement of atoms by high energy electron

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Inelastic scattering II

- Typical exposure requires 20 electrons/Å² on the specimen
- Equivalent to 10⁷ – 10⁸ rad in terms of radiation dose
- No biological specimen can survive this exposure
- Limited beneficial effect of lower specimen temperatures

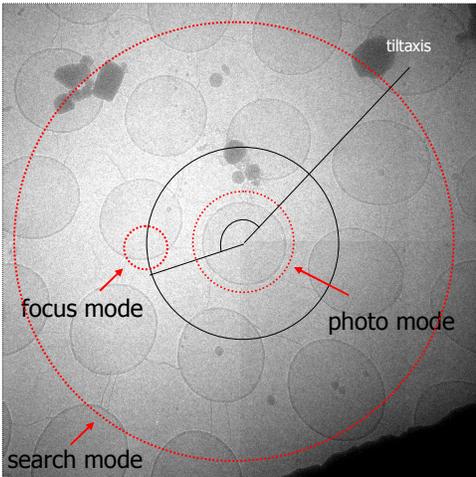
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Imaging Techniques in cryo-TEM

- Low intensity search mode, < 0.01 e/Å²/s
- Off-axis focus mode w. high intensity 50-100 e/Å²/s
- Record mode w. beam blanking on film/CCD

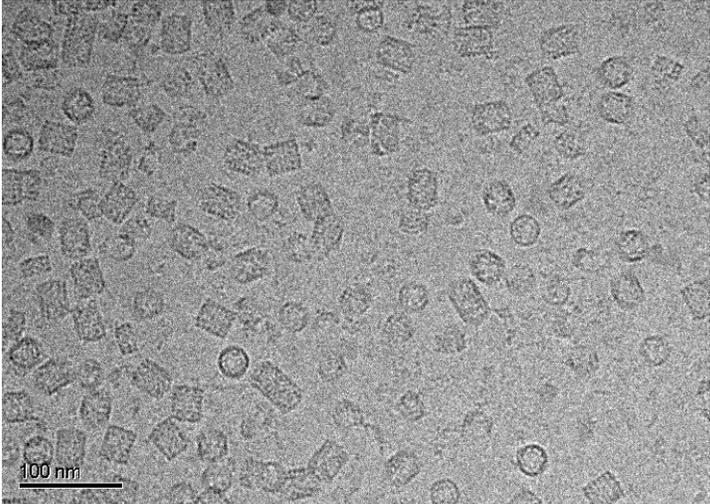


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3-7. Gusev + CULOS No
3-8. Gusev + CULOS

KLH in amorphous ice



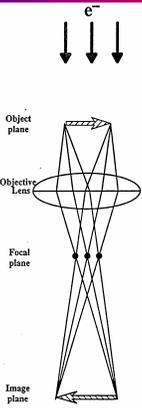
100 nm

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3-8. Gusev + CULOS

Wave Theory in EM



Object plane

Objective Lens

Focal plane

Image plane

e^-

Object Coulomb potential function $V(x_o, y_o, z_o)$

Object transmitted wave function $\Psi_o(x_o, y_o)$
 $\Psi_o(x_o, y_o) = 1 + i\sigma v(x_o, y_o)$
 $v(x_o, y_o) = \int V(x_o, y_o, z_o) dz_o$ ← Weak phase approx.

Phase shift $\gamma(S)$ introduced by objective lens
 $\gamma(S) = 2\pi(\frac{1}{4} C_s k^3 S^4 - \frac{1}{2} \Delta ZLS^2)$ ← Aberration function

Diffraction wave function $\Psi_d(S_x, S_y)$
 $\Psi_d(S_x, S_y) = F(S_x, S_y) \exp(i\gamma(S))$
 $F(S_x, S_y) = \mathcal{F}[\Psi_o(x_o, y_o)]$
 Diffraction intensity $I_d(S_x, S_y) = \Psi_d^*(S_x, S_y) \Psi_d(S_x, S_y)$ ← Diffraction description

Image wave function $\Psi(x_i, y_i)$
 $\Psi(x_i, y_i) = \mathcal{F}^{-1}[\Psi_d(S_x, S_y)]$
 Image intensity $I_i(x_i, y_i)$
 $I_i(x_i, y_i) = \delta(0, 0) - 2\sigma v(x_i, y_i) * \mathcal{F}^{-1}[\sin \gamma(S)]$ ← Image description

Computed diffraction wave function $T(S_x, S_y)$
 $T(S_x, S_y) = \mathcal{F}[I_i(x_i, y_i)]$
 $= \delta(0, 0) - 2 F(S_x, S_y) \sin \gamma(S)$

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What does a diff. pattern reveal?

$$T(S_x, S_y) \cdot T^*(S_x, S_y) = F^2(S_x, S_y) \cdot \sin^2 \gamma(S_x, S_y) \cdot E^2(S_x, S_y) + N^2(S_x, S_y)$$

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What does a diff. pattern reveal?

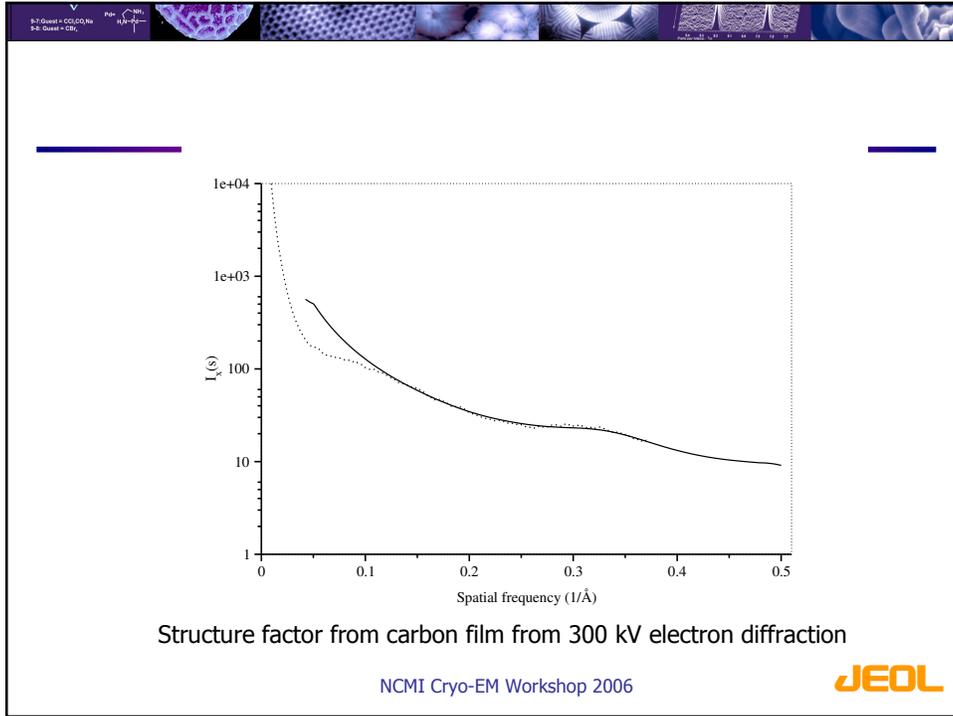
$$T(S_x, S_y) \cdot T^*(S_x, S_y) = F^2(S_x, S_y) \cdot \sin^2 \gamma(S_x, S_y) \cdot E^2(S_x, S_y) + N^2(S_x, S_y)$$

↓
Specimen's structure factor

Conveniently obtained from X-ray solution scattering data, e.g. 1D scattering plots of HSV-1, p22, or (with some pain) from electron diffraction patterns

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What does a diff. pattern reveal?

$$T(S_x, S_y) \bullet T^*(S_x, S_y) =$$

$$F^2(S_x, S_y) \bullet \sin^2 \gamma(S_x, S_y) \bullet E^2(S_x, S_y) + N^2(S_x, S_y)$$

Contrast Transfer Function

Oscillatory function describing the level of contrast of a particular spatial frequency

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What does a diff. pattern reveal?

$$T(S_x, S_y) \cdot T^*(S_x, S_y) = F^2(S_x, S_y) \cdot \sin^2 \gamma(S_x, S_y) \cdot E^2(S_x, S_y) + N^2(S_x, S_y)$$

Total Envelope Function

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Envelope Functions

- Decay in the power spectrum due to coherence limitations:
 - Spatial – \propto illumination angle
 - Temporal – \propto energy spread
- Can also be caused by “environmental factors”:
 - Drift from the holder
 - Acoustic coupling of the cryo-holder

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Envelope Functions II

- Gaussian type source:

$$G_{sc}(s) = \exp(-\pi^2 \alpha^2 (C_s \lambda^2 s^3 - \Delta Z s)^2)$$

- α dominates total envelope at $\Delta Z \geq 0.5 \mu\text{m}$

- Gaussian type fluctuations:

$$G_{tc}(s) = \exp\left(-\frac{\pi^2}{16 \ln 2} C_c^2 \lambda^2 \left(\frac{\Delta E}{E}\right)^2 s^4\right)$$

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Envelope Functions III

- Gaussian type fluctuations :

$$G_{oi}(s) = \exp\left(-\frac{\pi^2}{4 \ln 2} C_c^2 \lambda^2 \left(\frac{\Delta I}{I}\right)^2 s^4\right)$$

- Sinosoidal type fluctuations:

$$G_{lm}(s) = J_0(\pi \Delta f \lambda s^2)$$

- Drift:

$$G_m(s) = \frac{\sin(\pi s \Delta r)}{\pi s \Delta r}$$

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What does a diff. pattern reveal?

$$T(S_x, S_y) \cdot T^*(S_x, S_y) = F^2(S_x, S_y) \cdot \sin^2 \gamma(S_x, S_y) \cdot E^2(S_x, S_y) + N^2(S_x, S_y)$$

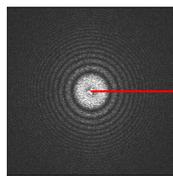
Total noise component

From a variety of sources (Gaussian, exponential, other). Some difference has been observed between energy-filtered and unfiltered data.

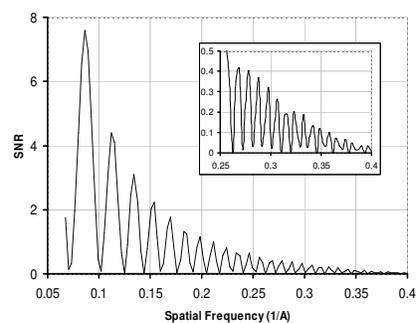
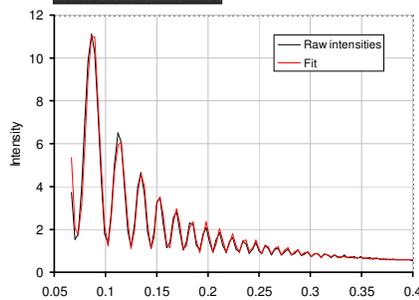
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Fitting the intensities

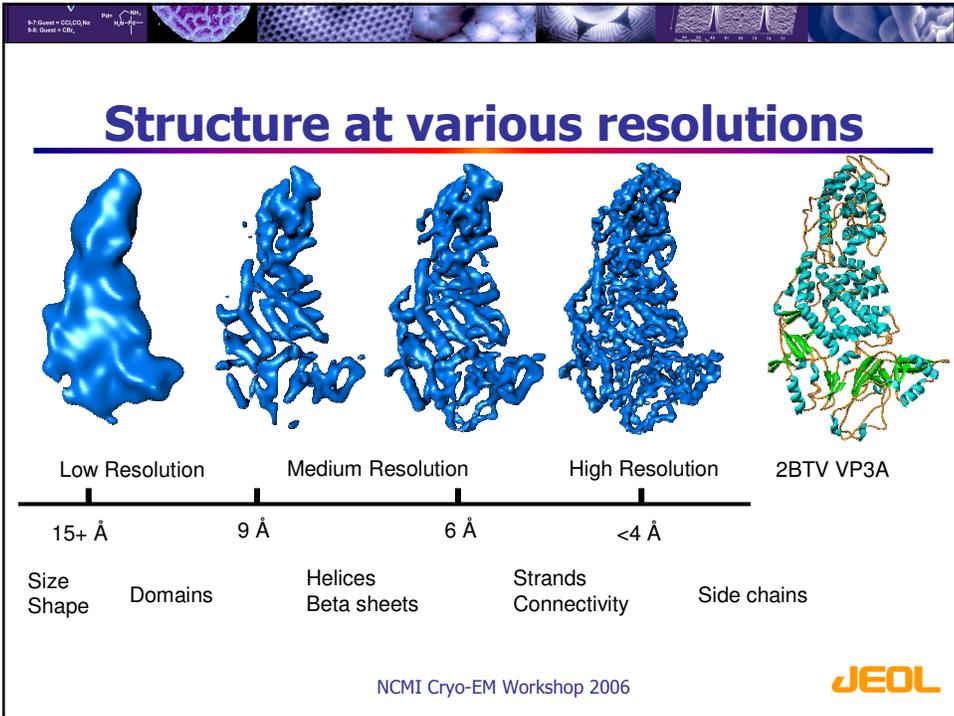
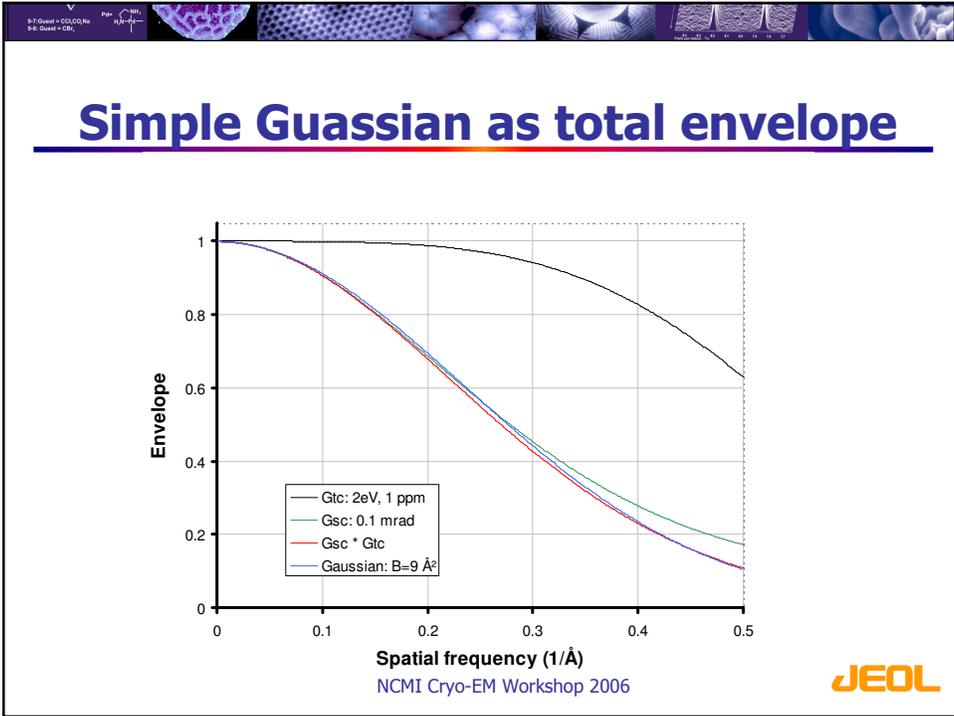


JEOL3000SFF, $\Delta Z = 0.97 \mu\text{m}$



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Resolution

The Theoretical Resolution Limit of the Electron Microscope

O. SCHERZER
Signal Corps Engineering Laboratories, Fort Monmouth, New Jersey
(Received June 14, 1948)

The resolving power of the electron microscope and the contrast in the image are calculated for different conditions of focusing, illumination and aperture. These conditions can change the limit of resolution by a factor of about 3. The contrast in the image of an atom is appreciably increased by defocusing and spherical aberration. Nevertheless, the contrast improves when the numerical value of the aberration constant is diminished. The effect of different methods of spherical correction is discussed briefly.



Appl. Physics (1949) 20, 20-29

Theorem: Chromatic and spherical aberrations are unavoidable for rotationally symmetric lenses. In addition, all aberrations add in quadrature, meaning one cannot combine lenses with suitable aberrations to cancel them in an attempt to pursue higher resolution.

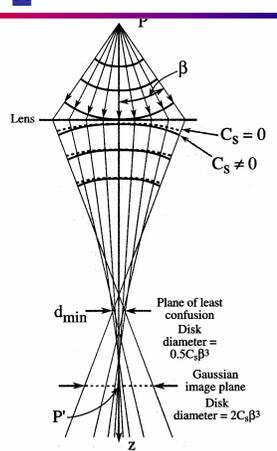
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Aberrations I

- Spherical aberration



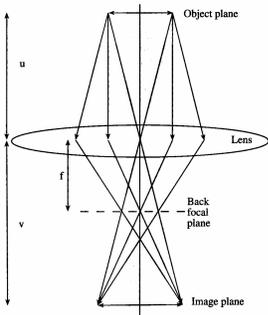


Figure 6.3. A complete ray diagram for a finite object, symmetrically positioned around the optic axis. All rays emerging from a point in the object (distance u from the lens) that are gathered by the lens converge to a point in the image (distance v from the lens) and all parallel rays are focused in the focal plane (distance f from the lens).

Figure 6.11. Spherical aberration in the lens causes wavefronts from a point object P to be spherically distorted. The point is thus imaged as a disk with a minimum radius in the plane of least confusion and a larger disk at P' in the Gaussian image plane.



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3-7. Guinier + CO₂O Na
3-8. Guinier + Cl₂

3-7. Guinier + CO₂O Na
3-8. Guinier + Cl₂

Aberrations II

- Chromatic aberration

Specimen
 β
Lens
Energy-loss electrons
No-loss electrons
Disc of least confusion
Gaussian image plane

Figure 6.12. Chromatic aberration results in electrons with a range of energies being focused in different planes. Electrons emerging from the specimen with no loss of energy are less strongly focused than those that suffered energy loss in the specimen, so a point is imaged as a disk.

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3-7. Guinier + CO₂O Na
3-8. Guinier + Cl₂

3-7. Guinier + CO₂O Na
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New directions

- Resolution limitations:
 - Lenses
 - Specimen

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New directions II

- Resolution limitations & goals:
 - Lenses – make them better
 - Specimen – make it less susceptible

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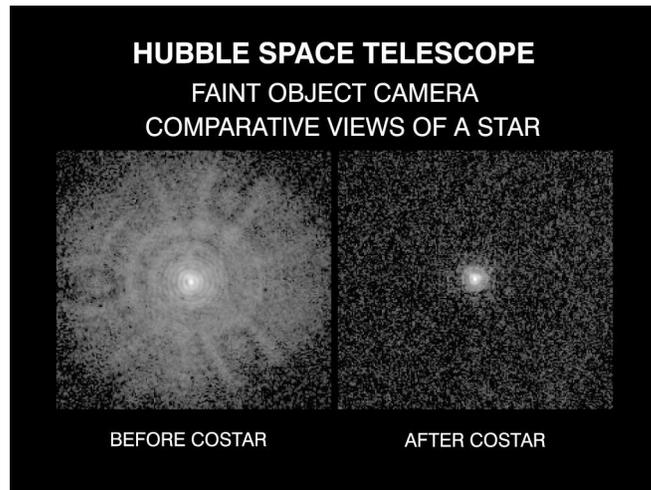
New directions III

- Resolution limitations & fixes:
 - Lenses – Cs correctors, phase plates
 - Specimen – spot-scan imaging, conductivity

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Are correctors useful for cryo-EM?



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C_s Correctors

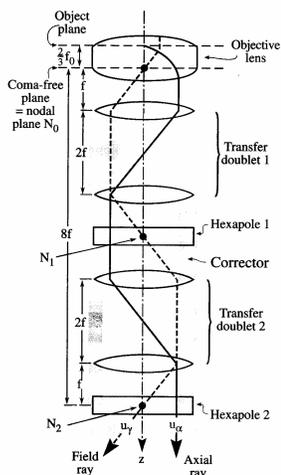
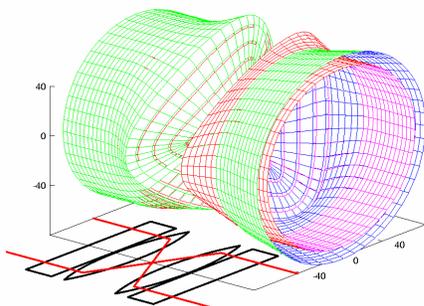
C_s correctors are used with TEM and STEM. There is a separate corrector for each imaging mode. The object of the corrector is to eliminate or greatly reduce spherical aberration.

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Principle of the Cs corrector

Principle of the Sextupole Corrector

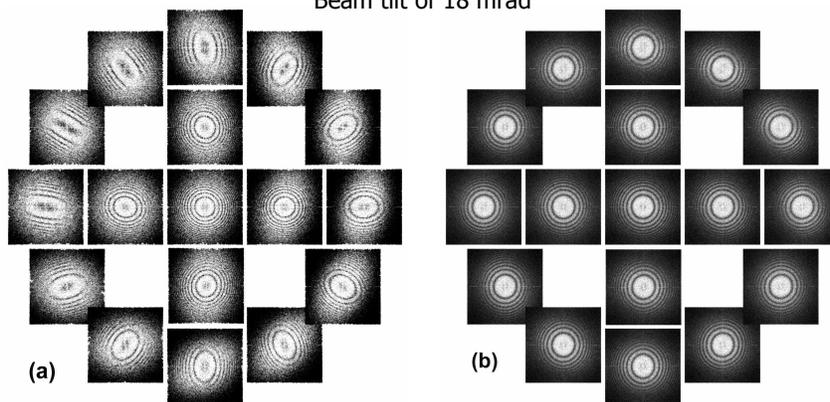


D.B. Williams & C.B. Carter,
Transmission Electron Microscopy (1996) pg. 469
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Effect of Cs correction

Beam tilt of 18 mrad



(a)

(b)

Uncorrected

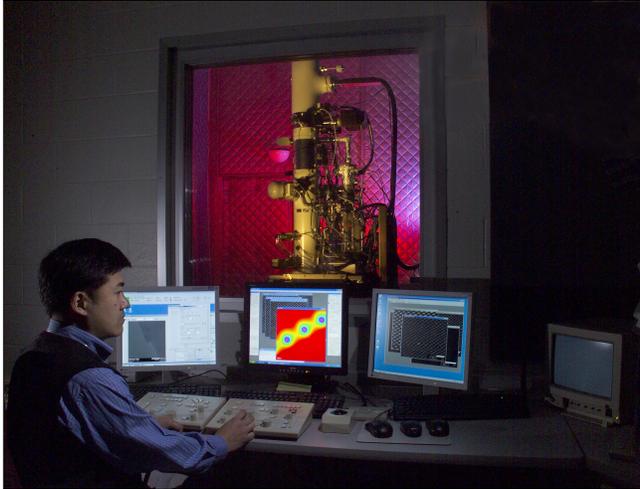
Corrected

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3-7. Goniol = CO_2O_2Na
3-8. Goniol = ClC

ACEM at ORNL



View from control room into instrument room

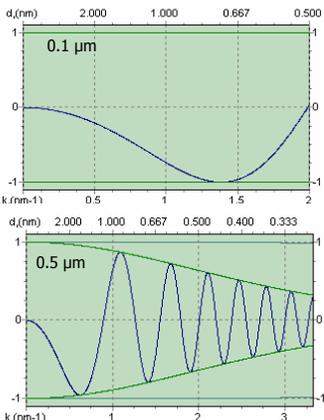
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3-7. Goniol = CO_2O_2Na
3-8. Goniol = ClC

Cs Correctors in Biology I

- Current interest focused on sub-10 Å.
- Specimen are low Z elements requiring large defocus for imaging.
- Extensive ringing in phase CTF



2100F; UHR p/p

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3-7. Guinier = CCl_2O_2 Na
3-8. Guinier = Cl_2

Cs Correctors in Biology II

- With Cs corrector the point resolution goes towards information limit.
- Also, larger p/p gap is possible thus allowing for environmental cells.
- No penalty for tilted illumination since coma is absent, thus allowing for “optical sectioning”
- Loss of contrast at low resolution can be compensated with phase plate.

point res. Info limit

$dF=14nm, Cs=0.5mm^{-1}$

$dF=14nm, Cs=0.005mm^{-1}$

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3-7. Guinier = CCl_2O_2 Na
3-8. Guinier = Cl_2

Different Phase Plates

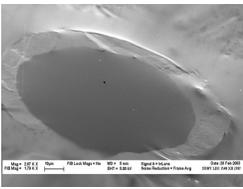
- Zernike phase plate – carbon film on BFP, small hole for the unscattered, direct beam
- Boersch phase plate or Einsellens – device in BFP w. floating inner ring at elevated potential
- Provides in-focus phase contrast, with increased transfer of lower spatial frequencies.
- Very suitable for frozen-hydrated specimens.

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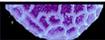
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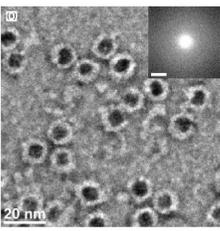
Zernike Phase Plate

Negatively-stained ferritin

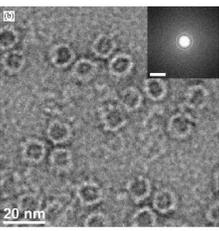


Phase plate

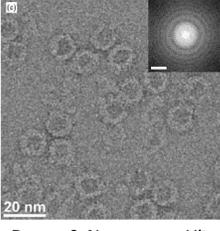




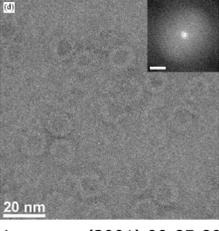
Conventional,
2.5 μm underfocus



Phase plate



Conventional,
0.5 μm underfocus



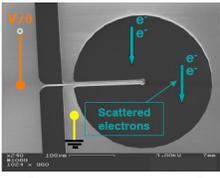
Conventional,
0.1 μm underfocus

Danov & Nagayama, Ultramicroscopy (2001) 90:85-89
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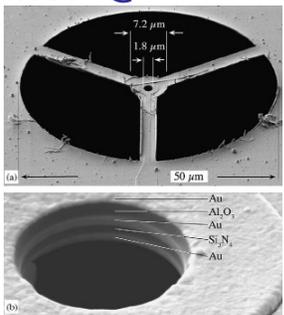


Boersch Phase Plate

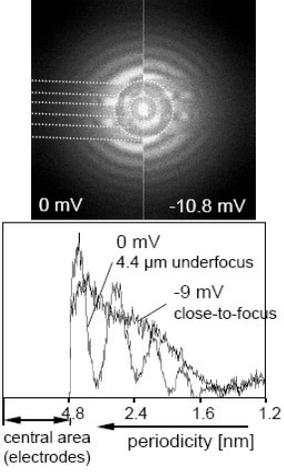
- Proposed in 1947
- Actively researched by Schröder @ MPG & Glaeser @ LBL



Glaeser



Schröder



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Specimen-related Limitations

- Charging:
 - SE emission
 - Very noticeable with tilted specimens
- Beam-induced specimen movement
 - Image amplitudes are only 1-10% of the theoretical values (Henderson & Glaeser)

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Specimen-related Limitations II

- Charging:
 - Adjust beam size carefully (Unwin)
 - Carbon-coat specimen (Glaeser, Brink)
 - BFP OLA
 - New support films, e.g. TiSi alloys
- Beam-induced specimen movement
 - Spot-scan imaging technique
 - Higher kV (Brink & Chiu)
 - Sturdier support films ?

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Acknowledgements

- Kersker, Isabell, Kawasaki, Armbruster
- Chiu lab
- Typke, Glaeser, Mansfield

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